System for Measuring the Phase Transition Temperature of KTa_{1-x}Nb_xO₃ Using Scanning Nonlinear Dielectric Microscopy

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Abstract

We have developed a system for measuring the paraelectric-to-ferroelectric phase transition temperature T_C using scanning nonlinear dielectric microscopy (SNDM). We developed a probe for samples with a huge dielectric constant and demonstrated that the T_C of potassium tantalate niobate (KTa_{1-x}Nb_xO₃, KTN) crystal can be measured locally using SNDM with this probe. The T_C measurement precision (standard deviation) is 0.09°C. This corresponds to a composition (Nb concentration *x*) resolution of 1.4×10^{-4} , which is difficult to achieve with other element analyzers. Moreover, by measuring T_C while changing the position, we demonstrated that we can measure the spatial distribution of the T_C of the KTN crystal. To reduce the time required for the operator to measure the T_C distribution of the KTN crystal, we developed software for fully automated operation.

1. Introduction

Potassium tantalate niobate (KTa_{1-x}Nb_xO₃, KTN) is a ferroelectric material and its dielectric constant and quadratic electro-optic (EO) constant (Kerr constant) are maximum around the paraelectric-to-ferroelectric phase transition temperature (T_C) [1]. These values are huge, so KTN is expected to reduce both the volume and driving voltage of EO devices, such as optical beam deflectors [2] and modulators [3]. In KTN, T_C can be adjusted by changing the composition, i.e., the Ta/Nb ratio [1]. T_C increases linearly with the amount of Nb (x) [4]. However, if there is a spatial change in the Ta/Nb ratio in a crystal, which may be induced by changes in the growth condition during crystal growth, T_C changes spatially, and the dielectric constant ε and the Kerr constant s change spatially at a given temperature. If a crystal's spatial distributions of ε and *s* are large (i.e., the crystal is highly inhomogeneous), it is difficult to guarantee the characteristics of an EO device manufactured from that crystal. Therefore, to guarantee the characteristics of a KTN device, it is necessary to evaluate the uniformity of a crystal's $T_{\rm C}$ before the device is manufactured. However, if $T_{\rm C}$ has to be estimated with a precision of 0.1°C, the composition should be estimated with a precision of 1.5×10^{-4} [4]. This is difficult to achieve with conventional element analyzers, such as an electron probe microanalyzer (EPMA).

In this paper, we describe the system for measuring the $T_{\rm C}$ of KTN that we developed. First, we review scanning nonlinear dielectric microscopy (SNDM) and describe the probe we developed for samples having a huge dielectric constant. Then, we describe our demonstration that the $T_{\rm C}$ of KTN crystal can be measured locally using SNDM with our probe. Experimental results for $T_{\rm C}$ measure-

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Fig. 1. Schematic diagram of SNDM.

ment precision and the spatial distribution of the $T_{\rm C}$ of the KTN crystal are also presented. Finally, we introduce the software developed to reduce the time taken for the operator to measure the $T_{\rm C}$ distribution of a KTN crystal.

2. Scanning nonlinear dielectric microscopy (SNDM)

2.1 Principle of SNDM

Scanning nonlinear dielectric microscopy (SNDM) has been developed to measure the local dielectric constant under a probe needle with high spatial resolution [5], [6]. Moreover, non-destructive measurement is possible, so a crystal that has been evaluated using SNDM can be used for manufacturing devices. A schematic diagram of SNDM is shown in **Fig. 1**. SNDM uses an LC (inductance and capacitance) oscillator with a probe needle. When the probe needle is far away from the sample, the resonant

frequency f_0 is $1/(2\pi\sqrt{LC_0})$, where L and C_0 are

the oscillator's inductance and capacitance (including stray capacitance), respectively. When the tip of the probe needle makes contact with the sample, the

resonant frequency f_S is $1/(2\pi\sqrt{L(C_0+C_s)})$, where C_S is the capacitance of the sample under the probe needle. As the dielectric constant of the sample

under the probe needle increases, C_S becomes larger and f_S becomes lower.

To check this, we measured the resonant frequency of several dielectric materials. The relationship between f_S and the dielectric constant at room temperature is shown in **Fig. 2**. This figure clearly indicates that the larger the dielectric constant, the lower f_S . Accordingly, when f_S is measured as a function of temperature, the local T_C can be obtained as the temperature that gives the lowest f_S . Moreover, the spatial distribution of T_C can be measured by changing the point touched by the probe.

2.2 Probe for keeping contact force constant

In crystals with a huge dielectric constant, such as KTN, f_S depends strongly on the contact force that the probe needle exerts on the sample. The reason for this is as follows: when the probe needle is in point contact with a sample having a huge dielectric constant, the electric field is concentrated in a very small area just under the tip of the probe needle [7], and f_S depends on the capacitance of that *very small* area. When the contact force is stronger, the contact area is larger due to deformation, and f_S depends on the capacitance of the *larger* area and is lower; namely, f_S depends on both the dielectric constant and the contact force.

In the work described here, we developed a probe that keeps the contact force constant during the mea-



Fig. 2. Resonant frequency f_S versus relative dielectric constant.



Fig. 3. Probe for keeping contact force constant. (a) Configuration, (b) no contact with sample, and (c) in contact with sample.

surement of $T_{\rm C}$. A schematic diagram of the probe is shown in **Fig. 3(a)**. The probe consists of a conductive tube and a conductive needle that can move up and down in the tube, which has a stopper to prevent the needle from falling out. When the tip of the probe is not in contact with a sample, the needle is held by the stopper, as shown in **Fig. 3(b)**. On the other hand, when the tip is in contact, the sample pushes the needle upwards, as shown in **Fig. 3(c)**. With this design, the contact force is equal to the force of gravity acting on the needle and can be kept constant.

3. Experimental results

3.1 Measurement of T_C of KTN

The resonant frequency f_S is plotted as a function of temperature *T* in **Fig. 4** when the probe needle (Au-coated, tip radius: 500 µm) came into contact with a KTN single crystal ($6 \times 5 \times 0.5 \text{ mm}^3$, $x \approx 0.4$). We estimated T_C to be about 35° C because domains appeared or disappeared at this temperature. The f_S was about 1214 MHz when the probe needle was kept far away from the sample; namely, $C_S = 0$. We used a Peltier device and a controller to sweep the temperature and we collected data during the cooling phase. The sweep rate was not constant, but the average rate was about 0.3° C/s. As shown in the figure, f_S was minimum at around 35° C. This indicates that SNDM can measure the T_C of KTN.

3.2 *T*_C measurement precision

We evaluated the precision of this method. We performed the $T_{\rm C}$ measurement N times, where N = 101. A histogram of $T_{\rm C}$ is shown in **Fig. 5**. Here, for $T_{\rm C}$, we used the temperature at which $df_{\rm S}/dT$ was the

T_c

35.5

34.5

1 mm

KTN crystal



Fig. 4. Resonant frequency f_s as a function of temperature when the probe made contact with KTN crystal.

5 mm



Fig. 5. Histogram of 101 measurements of T_C.

smallest because it changes drastically around $T_{\rm C}$, as shown in Fig. 4. This large change is induced by an abrupt change in the dielectric constant. The mean value of $T_{\rm C}$, \overline{T}_{C} , was 35.6°C, and the standard deviation $\sqrt{\sum_{i=1}^{N} (T_{Ci} - \overline{T}_{C})^2 / (N-1)}$ was 0.09°C. If we use the standard deviation as the precision, then the precision of this method is 0.09°C, which corresponds to a composition resolution of 1.4×10^{-4} , using the empirical equation $T_{\rm C} = 676x + 32$ measured in kelvin, where *x* is the amount of Nb [4]. This precision for the composition is difficult to achieve with other element analyzers, such as an EPMA.

3.3 Spatial distribution of $T_{\rm C}$

1 mm

Next, we evaluated the $T_{\rm C}$ distribution of the crystal. We measured $T_{\rm C}$ at 20 (= 5 × 4) points with a spacing of 1 mm. The distribution is shown in **Fig. 6**. For this crystal, $T_{\rm C}$ is higher in the top left of the figure. The maximum and minimum $T_{\rm C}$ values were 35.37 and 34.7°C, respectively. Here, $T_{\rm C}$ = 35.37°C corresponds to x = 0.4091 and 34.7°C to x = 0.4081, where x is the amount of Nb. Therefore, the variation in $T_{\rm C}$ in this crystal was 0.67°C, which corresponds to a variation in composition of about 0.001 using the above empirical equation [4].

Fig. 6. Spatial distribution of T_C of KTN crystal.

6 mm

4. Measurement system

To reduce the time required for the operator to evaluate the $T_{\rm C}$ distribution of the KTN crystal, we automated the measurement and data processing. A window of the software we developed is shown in **Fig. 7**. After the measurement condition (measurement spacing and number of points etc.) has been assigned and the sample has been set, the measurement is finished in a fully automated manner. Moreover, to eliminate manual data processing after the measurement, the software detects $T_{\rm C}$ by a differential calculation, as discussed in section 3.2 and records it.



Fig. 7. Window of software developed for fully automated measurement.

5. Conclusion

We described our system for measuring the phase transition temperature $T_{\rm C}$ of KTN using SNDM and a newly developed probe designed to keep the con-

tact force constant during the temperature sweep. The precision (standard deviation) is 0.09° C, which corresponds to a composition resolution of 1.4×10^{-4} . By measuring $T_{\rm C}$ while changing the position, we demonstrated that we could measure the $T_{\rm C}$ distribution of the KTN crystal. We also developed software capable of measuring the $T_{\rm C}$ distribution of the KTN crystal in a fully automated manner. This measurement system will enable the uniformity of KTN crystals to be evaluated, so it will be possible to guarantee the characteristics of KTN devices made from KTN crystals.

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